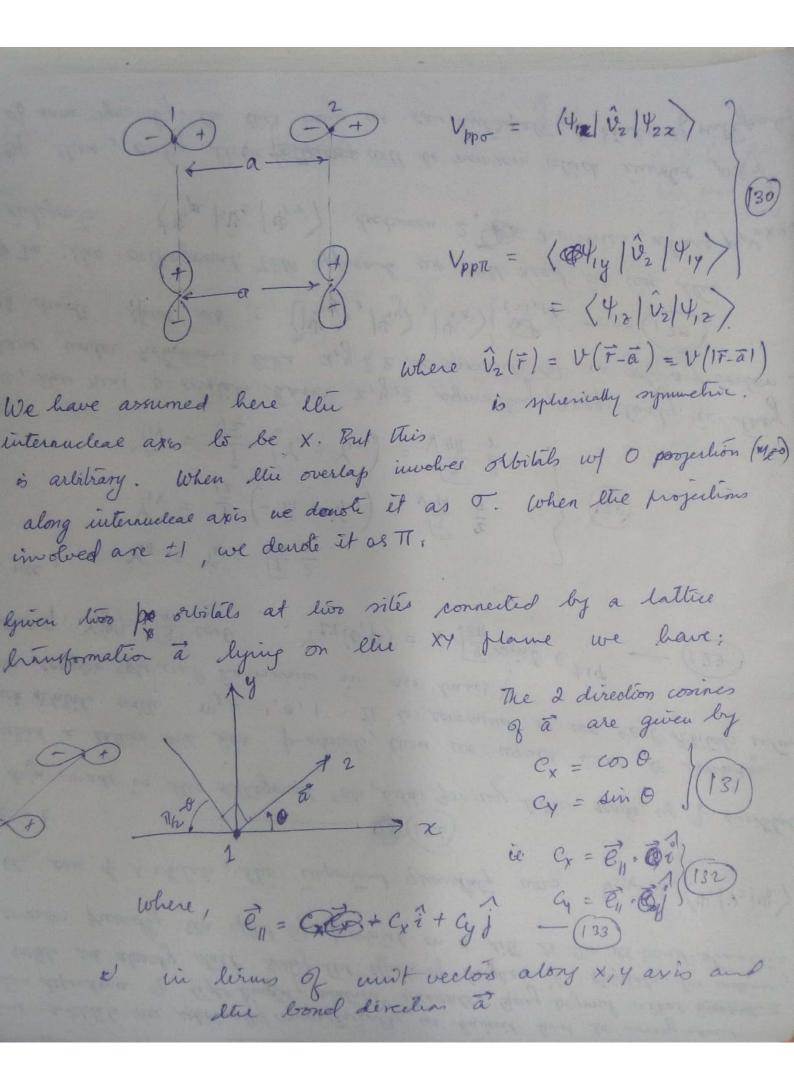
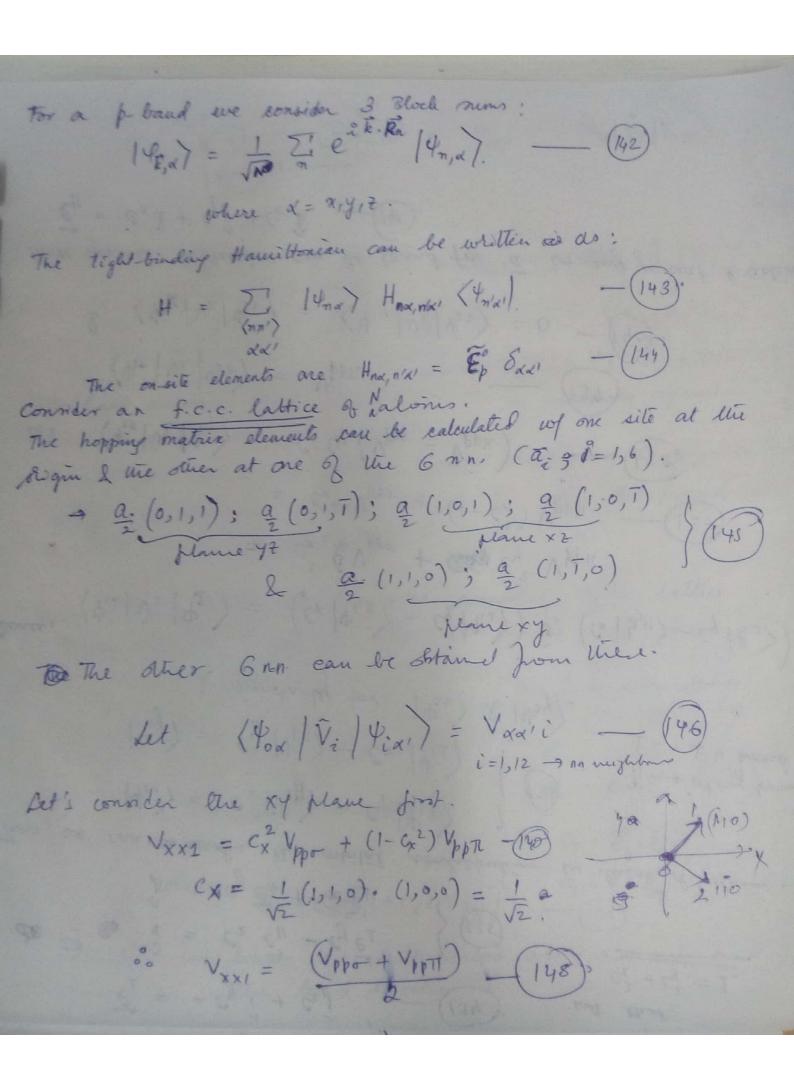
17/1/2018 f15/1/2018 Rec. 7-48 Tight-bruding with p-orbitals Since s-obitals are spherically symmetrically, we haven't had to worm about direction dependence of light-bringley matrix elements. Going beyond alkali tettative esta melab we already start seeing the effects of higher it orbitals. For instance, we'll consider presently the effect of p-colitals on a site to the to band-structure. In the case of s-orbitals, the important quantity was = -t (126) This defu. made in the orthogonal TBM, holds for any lattice made up of s-orthod, If instead a lattice site has ϕ -orbitals, then we would have to sonsider 3 such orbitals with $m_e = -1, 0, 1$. It is convarient to use real orbitals interest of the solution of the solut I the complex spherical harmonice in our basis. 80 instead of, $Y_{10}^{(99)} = \int_{4\pi}^{3} \cos \theta$, $Y_{\pm 1}(\theta, \varphi) = \pm \int_{8\pi}^{3} \sin \theta e^{\pm i\varphi}$ — (23) $Y_{12} = Y_{10} = \overline{A_{11}} \frac{2}{7}$ $Y_{1X} = \frac{1}{\sqrt{2}} \left(-Y_{11} + Y_{1-1} \right) = \overline{A_{11}} \frac{2}{7}$ $Y_{1Y} = \frac{1}{\sqrt{2}} \left(Y_{11} \pm Y_{1-1} \right) = \sqrt{\frac{3}{41}} \frac{4}{7}$ $Y_{1Y} = \sqrt{\frac{3}{41}} \frac{4}{7}$ Hence, the real p-obitals have x, y, 2 symmetry, respectively. i.e. they behave under rotations like x, y & 2, components of an ordinary vector. Let's denote these as : [Pig], | Piy], | Yiz] [2], N. — (129) to In the orthogonal TBM approach we will need to use the intégrals (4/1/2/4/22) between 2 De n-n estes, where M, V=x, y, Of these, only lies following will be non-zero which involve plan of same symmetry. On this basis we can auticipale I kinds quilipal.



$$\begin{array}{c} \vec{e_1} = -c_y \, \hat{i} + c_z \hat{j} & \qquad (3y) \\ \vec{e_2} = c_z \, \hat{i} + c_z \, \hat{e}_1 \\ \vec{f} = c_y \, \hat{e}_1 + c_x \, \hat{e}_2 \\ \vec{f} = c_y \, \hat{e}_1 + c_x \, \hat{e}_1 \\ \vec{f} = c_y \, \hat{e}_1 + c_x \, \hat{e}_2 \\ \vec{f} = c_y \, \hat{e}_1 + c_x \, \hat{e}_2 \\ \vec{f} = c_y \, \hat{e}_1 + c_x \, \hat{e}_1 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_1 - c_y \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 - c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 - c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 - c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 - c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 - c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f} = c_x \, | \vec{f}_1 | \vec{f}_2 \\ \vec{f}_1 | \vec{f}_2 \\ \vec{f}_1 | \vec{f}_2 \\ \vec{f}_1 | \vec{f}_2 | \vec{f}_2 \\ \vec{f}$$



$$V_{XYI} = C_X C_Y \left(V_{PP}\sigma^{-} - V_{PP}\Pi\right)$$

$$= \frac{1}{\sqrt{L}} \sqrt{L} \left(V_{PP}\sigma^{-} - V_{PP}\Pi\right) = \left(V_{PP}\sigma^{-} - V_{PP}\Pi\right) - \left(V_{PS}\right)$$

$$V_{ZZI} = V_{PP}\Pi$$

$$V_{XZI} = \frac{1}{\sqrt{L}} \left(V_{P}\sigma^{-} - V_{PP}\Pi\right) = \left(V_{PP}\sigma^{-} - V_{PP}\Pi\right) - \left(V_{PS}\right)$$

$$V_{XZI} = \frac{1}{\sqrt{L}} \left(V_{P}\sigma^{-} - V_{PP}\Pi\right) = \left(V_{PP}\sigma^{-} - V_{PP}\Pi\right) - \left(V_{PS}\right)$$

$$V_{XZI} = V_{PP}\Pi$$

$$V_{YZI} = V_{$$

$$H_{xx}(\vec{k}) = \tilde{\epsilon}_{p}^{o} + e^{\frac{i(ky+kp)a}{2}} V(0,1,1)_{x,x} + e^{-\frac{i(ky+kp)a}{2}} V(0,1,1)_{x,x}$$

$$+ e^{\frac{i(ky+kp)a}{2}} V(0,1,1)_{x,x} + e^{-\frac{i(ky+kp)a}{2}} V(0,1,1)_{x,x}$$

$$+ e^{\frac{i(ky+kp)a}{2}} V(1,0,1)_{x,x} + e^{-\frac{i(ky+kp)a}{2}} V(1,0,1)_{x,x}$$

$$+ e^{\frac{i(ky+kp)a}{2}} V(1,1,1,0)_{x,x} + e^{-\frac{i(ky+kp)a}{2}} V(1,1,1,0)_{x,x}$$

$$+ e^{\frac{i(ky+kp)a}{2}} V(1,1,1,0)_{x,x} + e^{-\frac{i(ky+kp)a}{2}} V(1,1,0,1)_{x,x}$$

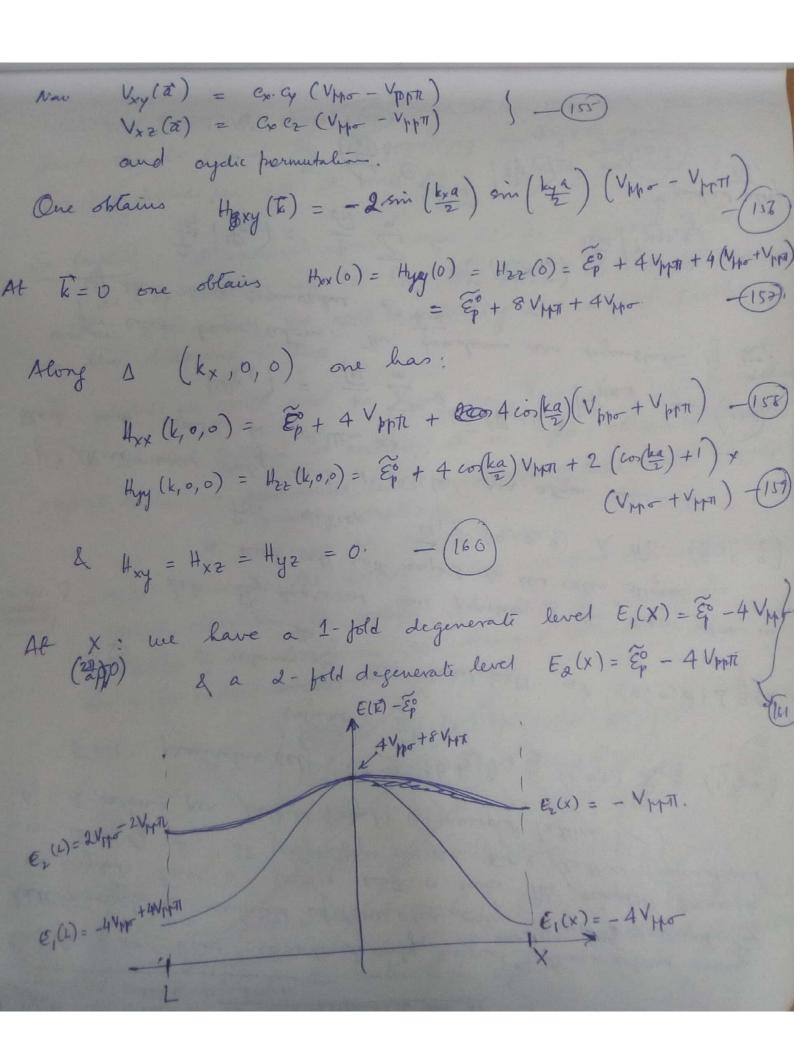
$$+ e^{\frac{i(ky+kp)a}{2}} V(1,1,1,0)_{x,x} + e^{-\frac{i(ky+kp)a}{2}} V(1,1,0,1)_{x,x}$$

$$+ e^{\frac{i(ky+kp)a}{2}} V(1,1,0,1)_{x,x} + e^{-\frac{i(ky+kp)a}{2}} V(1,1,0,1)_{x,x}$$

$$+ e^{\frac{i(ky+kp)a}{2}} V(1,1,0,1)_{x,x} + e^{-\frac{i(ky+kp)a}{2}} V(1,0,1)_{x,x}$$

$$+ e^{\frac{i(ky+kp)a}{2}} V(1,0,1)_{x,x} + e^{\frac{i(ky+kp)a}{2}} V(1,0,1)_{x,x}$$

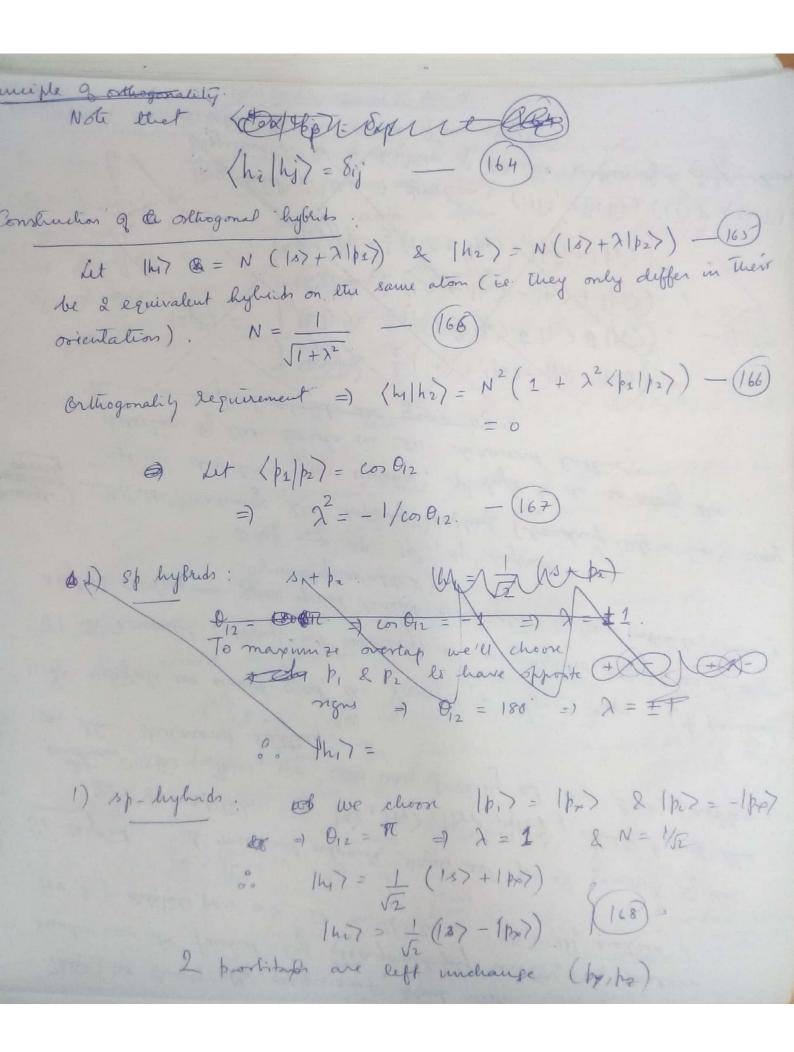
$$+ e^{\frac{i(ky+kp)a}{2}} V(1,0,1)_{x,x} + e^{\frac{i(ky+kp)a}{2$$

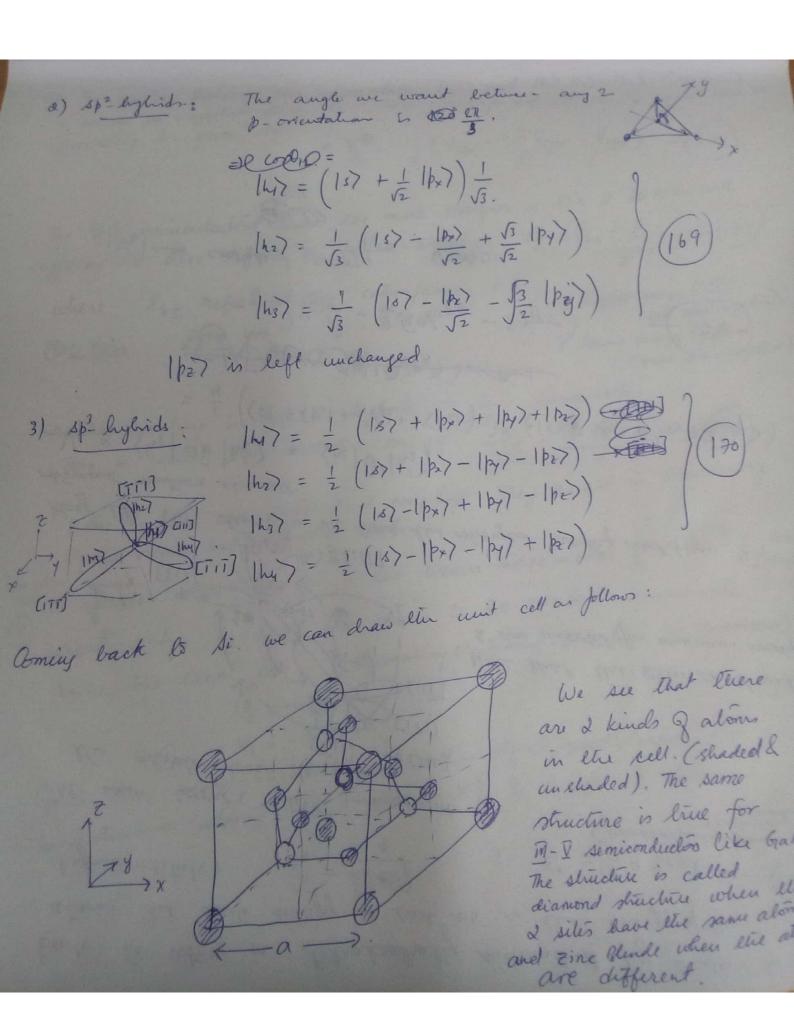


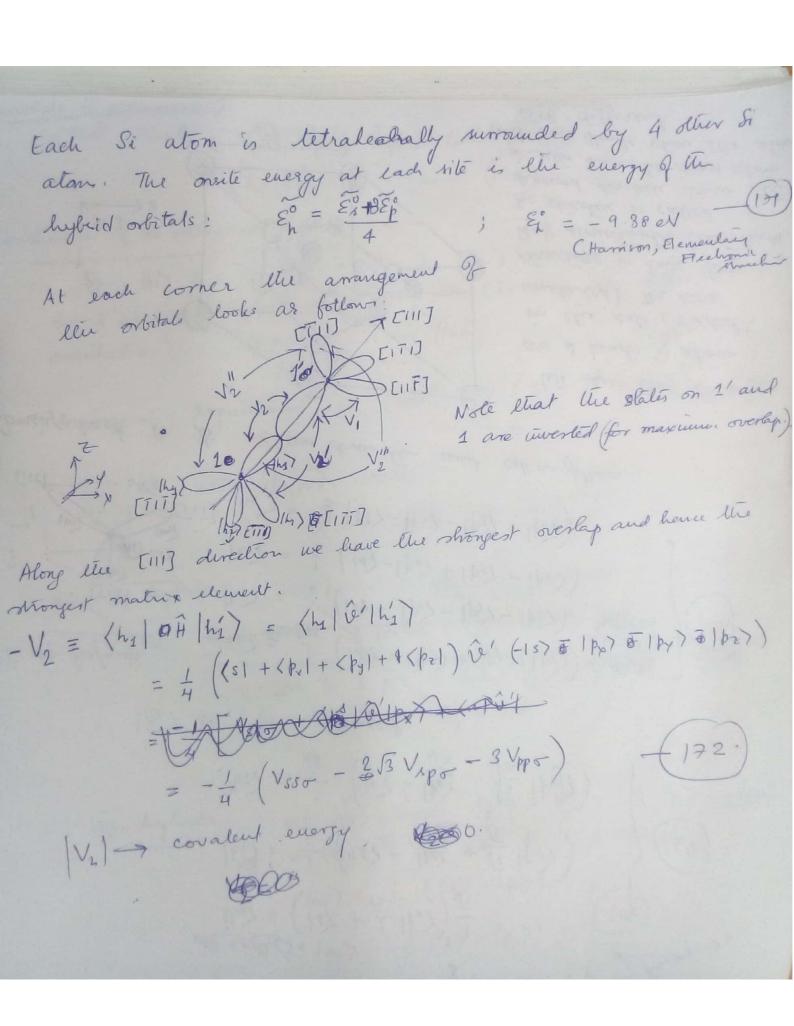
Lee 9. Tight-binding approach for Si withe a basis So for we have coundered only Bravais lattices where every lattice fourt is associated up one atom only, "Most" a semiconductor is that of si. Si crystallyes in an F.C.C. lattice formed out of I alone per lattice point. Diamond Lettice F.C.C promitive cell: 2 (0,1,1), 2 (1,0,1) & a (1,1,0) $volume = \frac{a^3}{4}$ Si: 2 atoms in the mit cell: at (0,0,0) & a (1) Let Un represent the position of the basis atoms in a unit cell with respect to the cells digin. for Silicon $\vec{u}_1 = (0,0,6)$ ($\vec{u}_2 = (\frac{a}{4}, \frac{q}{4}, \frac{q}{4})$ If there were only on I I orbital per atom then the Black sum would be walten as: 19 = 1 2 eik. (Rn + vi,) 14ne? - (16) for each basis atom. The functions are eigenstale of Ta, we the same eigenvalue ei k.a. (元) (年) = 一元 では、(日、 + 4) (日、 e = 1. N. = 200 e = 19E, RT (62) a. Let e = 1, No. -> (163)

Thus, we have No degenerate Bloch states (for Ta). The resulting bands can be found by diagonalizing an ON, × N, matrix. For 3 porbilats per site it will be a 3N, × 3N, matrix. Si layetel. A minimal atomic basis sel for Si consists of the 3s & 3p orbitals -> |87, |px7, |px7, |px7, |px7, centred ont the atomic sites. Thus, we tolk are looking at an 8x8 problem. The integrals we would need as Vsso, Vspo, Vppo, Vppo, Vppo (4 parameter It is convenient (although not necessary) to use hybrid orbitals as as bains. A trybud orbitals -> symmetry adapted). A & p -> Ap, sp2, sp3 hybrids. - Hybrids maximize the hoffing coupling (bonding interaction) along special directions. For instance, Sp3 hybrids do so along the directions of live bonds in the diamond lattice. The 4 sp3 higherd orbitals are given by. 15p3 = 1 (13) = 1 (15) + 1px) + 1p2) - (63).

15p3 = 1 (15) + 1px) + 1p2) - (163). They point in the 4 direction (111), (TII), (TII) & (III) & (IIII) & (III







The coupling between the along state (hi) and (hi) or (hy) (or v.v.) is smaller in magnitude that V_2 . These couplings are all exactly equal due is premisely & given by: $V_2 = -\frac{1}{4} \left(V_{550} - \frac{2\sqrt{3}}{3} V_{5}p_0 + V_{pp_0} \right)$ (173) for W. R semiconductors since the mate energies on the 2 to sites will expers, we also define a polar energy: $V_3 = E_{h+} - E_{h-} - (174)$ where $\mathcal{E}_{k\pm}$ refers to the cation/action site.

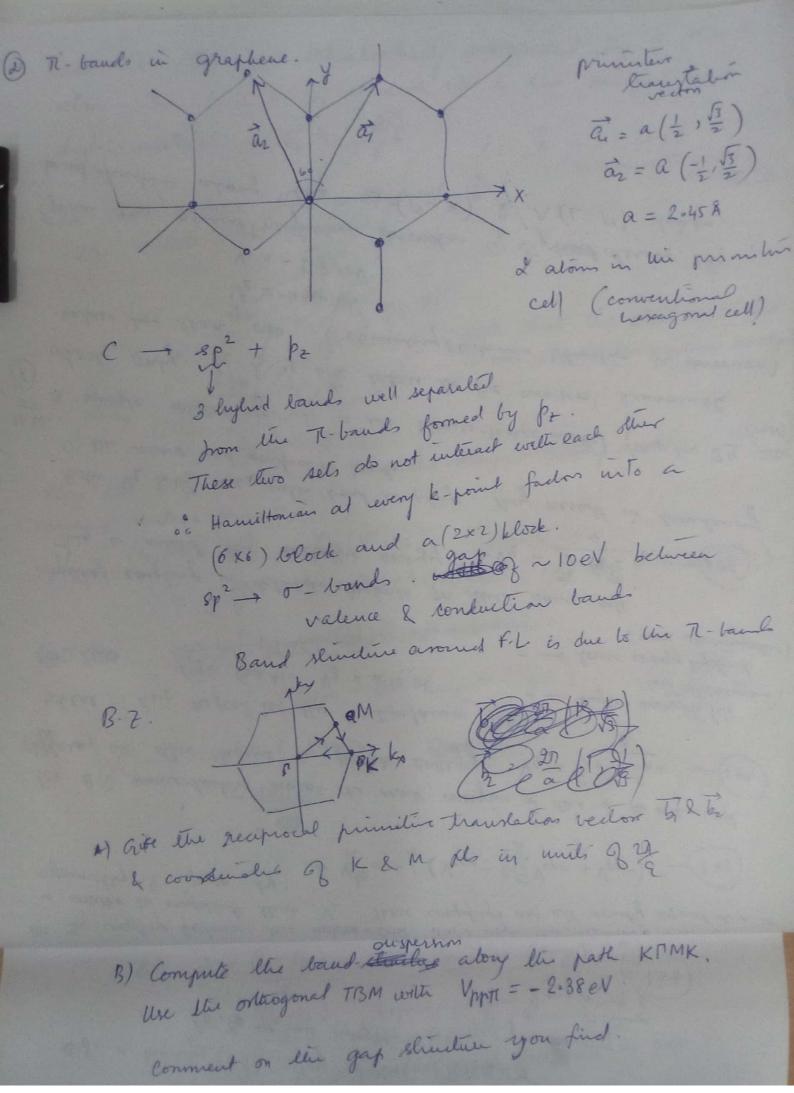
+ shigher energy hybrid
(less electrongalia)

- lower energy hybrid

- - lower energy hybrid - - lower every hydrid (more electronique) Another coupling is between hybrids on the same site.

This is called the metallic energy. $-V_1 = \frac{\mathcal{E}_p^2 - \mathcal{E}_q^2\mathcal{E}_q^2}{4}$ — (176). Both V2 & V' are inter-bond couplings. They gresult in broadening.

Quete valence and conduction bands. +W. imple model for si is the Wearn-Thospe model (Phys. Rev. 134, 2508 (1971)) where only V2 & V1 are laken to be non-zero. Reasonable values for then are (Elementary Electronic Structure, W. Harrison) V2 = -4.44 eV V1 = - 1.80eV Given the nearest neighbour corvectors of Si, and derive the band structure along the D. (T-X) & N(L-T) lines.



Lec 10. Systems with bandgap: Reach Distortion Pld Bring Alley 4/1/2019 Show Si band-structure (from A.P. Sutton). -> 4 & valence bands completely occupied → VB maximum at I and EB minimum at X (in actuality) -> Band gap a (smallest range of forbidden inergies) is 101eV and of indirect nature (1 -> x). (() () = P() = eit. 7 4(1) B P(F) = @1 BP(F) = (k + 10) P(F) In scattering processes in the crystal total momentum is conserved. In scattering processes in the compensated by appropriate of any charge in it should be compensated by appropriate The first state of the state of Optical proferties of si are affected by the presence of undirent Inclusion of d'orbitals in the Ao basis improves the description of conduction band, and gaps can be realized with more that & I orbital per atom as in Ri, or even by more than one atom in the unit cell,

Lopping ultigral; 177

-t = (4/4/A/4/B) Let $\Delta = (\varepsilon_A - \varepsilon_B)$ $\varepsilon = (\varepsilon_A + \varepsilon_B)$ HW. O Calculate the band structure and determine the gap in lerms of t, 1, E. (2) Esterolate live total density of states to D(E). 3 Plot the local density of states $\mathcal{D}_A(E) & \mathcal{D}_B(E)$. 4) For algira Given ettat each site A/B contibutes 1 e, delerative lu average charge difference between A & B sublittice per unit cell. (5) Hors would the above go charge difference depend on the Fermi energy? Density of states: when the nystem becomes very large, many energy levels nesult and they are distributed in a dense Jashim. with near In a given energy hange many energy state can be packed. and their number To see this hange many energy state can be packed. and their number of alons as before. let's first consider. the infinite linears chain of hi alons as before. let's first consider the infinite linears chain of hi alons as before. The number of he quantities numbers allowed = leagth of B.7 = No. 8 unit passing k-pts. co number of states per unit length of B.Z. = $\frac{Na}{2\pi/a} = \frac{Na}{3\pi} = d(k)$. What is the number of states per unit energy interval to placed atound a certain energy E? Certain everyy E?

Let Mor number of states upto energy & from the bottom most part Then the density of states is defined through? $D(E) = \frac{dN}{dE} = \frac{dN}{dk} \left| \frac{dk}{dE} \right| = \frac{dN}{dk}$ Excelled 1st describer The modulus of the is taken because D(E)

cannot be negative. The facts of 2 accounts for the ±k degenerary of Ell). for a 1-d linear chair of hi atom. we can evaluate this

by noting that $dN = d(k) = \frac{Na}{2\pi}$ Let $dk = \frac{Na}{2} = \frac{1}{2}$ Let $dk = \frac{Na}{2} = \frac{1}{2}$ $= \left| \frac{d\ell}{d\ell} \right| = 2ta \sin(ka) - (181)$ 00 D(E) = 2 Noa x _ 1 _ (182) $= \frac{800}{071} \frac{1}{\sqrt{4t^2 - (\varepsilon - \varepsilon_0)^2}} - \frac{183}{\sqrt{14t^2 - (\varepsilon - \varepsilon_0)^2}}$ Frat E 30152t

If we integrale over all the energies in the band we get 1. $\widetilde{\mathcal{E}}_{0} = 2 \text{ st} \qquad -(84)$ This barically means that there is I state per atom. Often the doors is used to mean the density of occupied states. In this case it is customary to multiply (183) who a factor of 2 for double occupany of each level and also occupation function f(E). Es. for a syclim at And a doi per atom o (or per unit volume) is defined as; OKOK, $f(E) = 2\Theta(E - E_f)$ — (85). $d(E) = \frac{1}{N} D(E) = \frac{1}{\pi} \frac{1}{\sqrt{4t^2 - (E-E)^2}} - \frac{1}{(86)^2}$ For an infinite chairs since all sites are equibalent Nd(E) = D(E). But for systems with defects of edges of enequivalent atoms (AB alloy)
this neddation is not line. the To dislinguish "bulk" and "surface" or between sublattices we can define a local density of states . (LDO). for this eigenstate probability of finding an election coord the atom site on is: P(E) = |(4/n | 4/E)|^2 - (88).

The energy we can use P(E) to weight the DOS. to determine the energy two can use P(E) to weight 14/m? If there are every from the atomic state 14m? If there are every from the atomic state 14m? If there are still do.s. arising from the atomic state 14m? If there are suite of the atomic state 14m? Nav, seice 1927 = [147 (4/1927 - (89)) dier Biguthalis with erergy E = E(E) lien we must num over there as well to get him total weight. Contribution of 142) to D(E) - 9 & 8(E-EME)) - (B)

